Multiphysics Degradation Modeling of Energy Storage Materials via RKPM with a Neural Network-Enhancement

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Li-ion Battery Electrode Microstructures and Electro-Chemo-Mechanical Cracking

Electrode Microstructure and **Electro-Chemo-Mechanical Cracking**

Cathode Composition:

- Randomly-oriented grains
- Anisotropic grain material properties



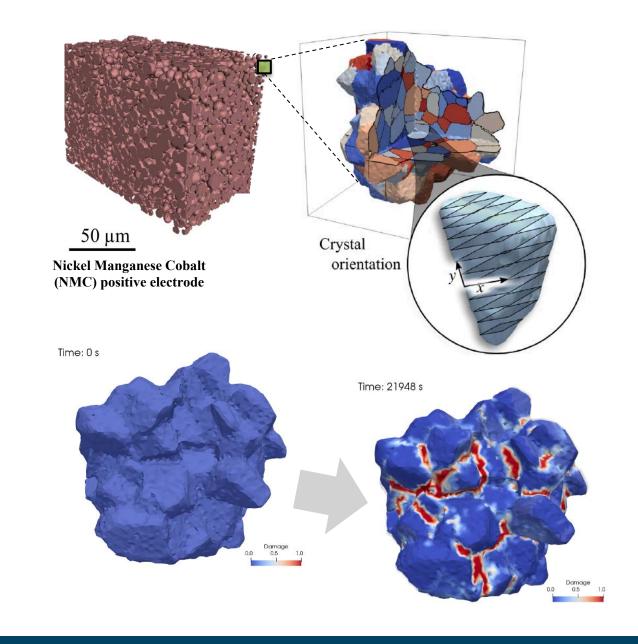
Charge Cycling:

Lithium movement between electrodes causes nonuniform grain expansion and contraction



Electro-chemo-mechanical cracking:

- Inhibited lithium flow via tortuous diffusion path
- Reduced battery life



^{2.} Allen, J., P. Weddle, A. Verma, et al. 2021. "Quantifying the influence of charge rate and cathode-particle architectures on degradation of Li-ion cells through 3D continuum-level damage models." J. Power Sources.



^{1.} NREL. "Battery Microstructures Library." https://www.nrel.gov/transportation/microstructure.html.

Governing Equations

Electrochemical

C

Lithium transport balance \rightarrow lithium concentration c

$$\dot{c} + \nabla \cdot \mathbf{J} = 0 \quad in \Omega$$

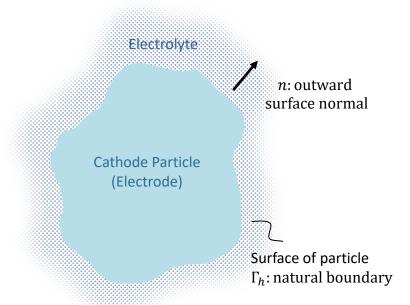
$$\mathbf{J} = -\mathbf{D} \cdot \nabla c$$

$$\mathbf{D} \cdot \nabla c \cdot \mathbf{n} = -\frac{\bar{J}(c, \Phi)}{F} \quad on \Gamma_{h_c}$$

Model

Ф

Solid-phase electrostatic potential balance \rightarrow potential Φ



Mechanical

 \boldsymbol{u}

Diffusion-induced mechanical deformation \rightarrow displacement u

$$abla \cdot \boldsymbol{\sigma} = \mathbf{0} \quad in \ \Omega$$
 $\boldsymbol{\sigma} = \mathbb{C}\boldsymbol{\epsilon}^{e}$
 $\boldsymbol{\epsilon}^{D} = \boldsymbol{\beta}\Delta C$
 $\boldsymbol{\sigma} \cdot \boldsymbol{n} = 0 \quad on \ \Gamma_{h_{1}}$

Butler-Volmer interface condition

$$\bar{J}(c, \Phi) = \bar{J}_0 \left[\exp\left(\frac{\alpha_a \eta F}{RT}\right) - \exp\left(-\frac{\alpha_c \eta F}{RT}\right) \right]
\eta(c, \Phi) = \Phi - \Phi_{el} - E^{eq} \left(\frac{c}{c_{max}}\right)$$

^{4.} G.L. Plett, Battery Management Systems, Volume I: Battery Modeling, Artech House, 2015.

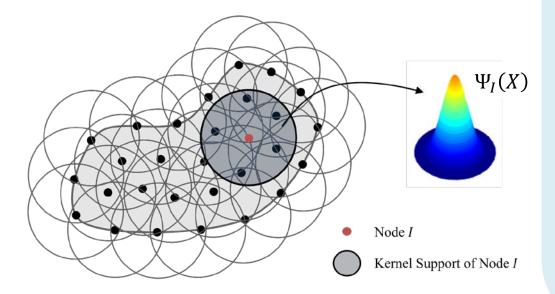
^{5.} Doyle, M., T. Fuller, J. Newman, "Modeling of Galvanistic Charge and Discharge of the Lithium/Polymer/Insertion Cell", Journal of the Electrochemical Society. 140 (1993). 1 6. Richardson, G.W., J.M. Foster, R. Ranom, C.P. Please, A.M. Ramos, "Charge transport modelling of Lithium-ion batteries", Eur. J. Appl. Math. 33 (2022).

Reproducing Kernel Particle Method (RKPM)

Reproducing Kernel (RK) Approximation

RK Approximation:

$$u(\mathbf{x}) \approx u^h(\mathbf{x}) = \sum_{I=1}^{NP} \Psi_I(\mathbf{x}) d_I$$



Shape Function Construction: $\Psi_I(x)$

Strategic Correction of Kernel Functions, ϕ_a :

$$\Psi_I(\mathbf{x}) = C(\mathbf{x}; \mathbf{x} - \mathbf{x}_I)\phi_a(\mathbf{x} - \mathbf{x}_I) = \left(\sum_{|\alpha| \le n} (\mathbf{x} - \mathbf{x}_I)^{\alpha} b_{\alpha}(\mathbf{x})\right) \phi_a(\mathbf{x} - \mathbf{x}_I)$$

$$\Psi_I(x) \equiv H^T(x-x_I) b(x) \phi_a(x-x_I)$$

controls order of ensures controls order of completeness satisfaction of continuity

reproducing conditions

$$\mathbf{H}^{T}(\mathbf{x}-\mathbf{x}_{I}) = [1, (x_{1}-x_{1I}), (x_{2}-x_{2I}), (x_{3}-x_{3I}), ..., (x_{3}-x_{3I})^{n}]$$

$$b(x) = M^{-1}(x)H(0)$$
, where $M(x) = \sum_{l=1}^{NP} H(x - x_l)H^{T}(x - x_l)\phi_a(x - x_l)$

Patch Test for Coupled Electro-Chemo-Mechanical Problem

Coupled Linear Patch Test Construction

Let's define a coupled BVP where the source term and BCs are associated with predefined fields.



The solution fields in the domain interior are expected to reproduce the predefined fields.

Predefined Fields:

$$c^p = 34720 + 2480x_1 + 7440x_2$$

$$\Phi^p = 3.8285 - 0.05x_1 - 0.1x_2$$

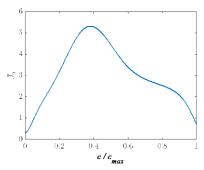
$$u_1^p = 10^{-1} x_1$$

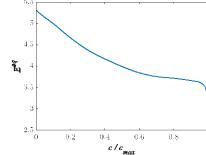
$$u_2^p = 2x10^{-1} x_2$$

Butler-Volmer interface condition

$$\bar{J}(c,\Phi) = \bar{J}_0 \left[\exp\left(\frac{\alpha_a \eta F}{RT}\right) - \exp\left(-\frac{\alpha_c \eta F}{RT}\right) \right] \quad on \ \Gamma_h$$

$$\eta(c,\Phi) = \Phi - \Phi_{el} - E^{eq} \left(\frac{c}{c_{max}}\right)$$





Governing equations:

$$\nabla \cdot (-\boldsymbol{D} \cdot \nabla c) = s^c \quad in \Omega$$

$$\nabla \cdot (\kappa \nabla \Phi) = s^{\Phi} \qquad in \Omega$$

$$\nabla \cdot \boldsymbol{\sigma} = \boldsymbol{s}^u$$
 in Ω

Boundary conditions:

$$D \nabla c \cdot \boldsymbol{n} = D \nabla c^{\boldsymbol{p}} \cdot \boldsymbol{n} + \frac{\bar{J}^{p}(c^{\boldsymbol{p}}, \Phi^{\boldsymbol{p}})}{F} - \frac{\bar{J}}{F} \quad on \ \Gamma_{h_{c}} = \Gamma$$

$$\kappa \nabla \Phi \cdot \boldsymbol{n} = \kappa \nabla \Phi^{p} \cdot \boldsymbol{n} + \bar{J}^{p}(c^{p}, \Phi^{p}) - \bar{J} \quad on \Gamma_{h_{\Phi}} = \Gamma$$

$$\boldsymbol{\sigma} \cdot \boldsymbol{n} = 0$$
 on $\Gamma_{h_u} = \Gamma$

Source terms:
$$s^c = 0$$
; $s^{\phi} = 0$; $s^u = 0$

Linear Patch Test Results for Coupled Electro-Chemo-Mechanical Problem

Relative Error Norm Equations

L_2 Norm:

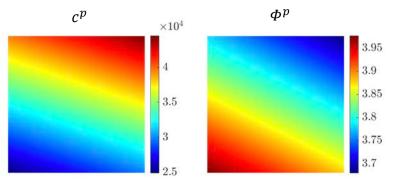
$$\| f - f^h \|_0 = \frac{\left(\int_{\Omega} (f - f^h)^2 d\Omega \right)^{\frac{1}{2}}}{\left(\int_{\Omega} f^2 d\Omega \right)^{\frac{1}{2}}}$$

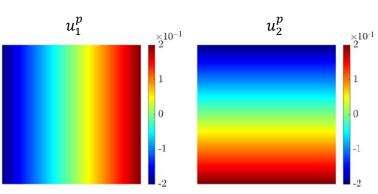
$$\| f - f^h \|_0 = \frac{\left(\int_{\Omega} (f_i - f_i^h)^2 d\Omega \right)^{\frac{1}{2}}}{\left(\int_{\Omega} f_i^2 d\Omega \right)^{\frac{1}{2}}}$$

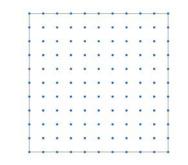
$$\parallel \boldsymbol{f} - \boldsymbol{f}^h \parallel_0 = \frac{\left(\int_{\Omega} (f_i - f_i^h)^2 d\Omega\right)^{\frac{1}{2}}}{\left(\int_{\Omega} f_i^2 d\Omega\right)^{\frac{1}{2}}}$$

H^1 Semi-Norm:

$$|f - f^{h}|_{1} = \frac{\left(\int_{\Omega} (f_{,i} - f_{,i}^{h})^{2} d\Omega\right)^{\frac{1}{2}}}{\left(\int_{\Omega} f_{,i}^{2} d\Omega\right)^{\frac{1}{2}}}$$
$$|f - f^{h}|_{1} = \frac{\left(\int_{\Omega} (f_{i,j} - f_{i,j}^{h})^{2} d\Omega\right)^{\frac{1}{2}}}{\left(\int_{\Omega} f_{i,j}^{2} d\Omega\right)^{\frac{1}{2}}}$$







Note: Eigenvalue shifting used to suppress rigid modes

Field	Relative Error L ₂	Relative Error H ¹
	Norm	Semi-norm
С	$1.23x10^{-14}$	$1.29x10^{-13}$
Φ	$5.96x10^{-15}$	$5.05x10^{-13}$
u	$8.21x10^{-11}$	$8.02x10^{-10}$

Mesh Convergence Study of High-Order Manufactured Solution

Mesh Convergence Study of High-Order Solution

Manufactured Fields:

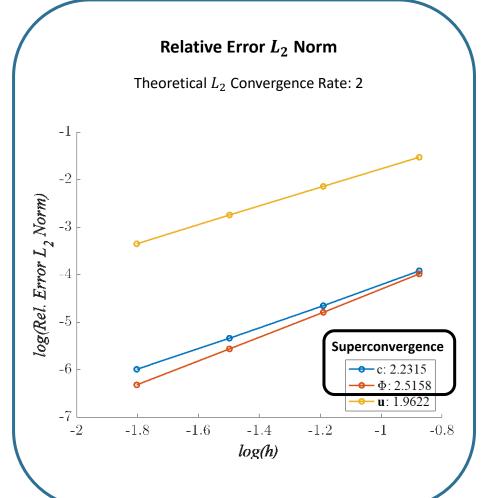
$$c^p = a_0 + A^c cos(i\pi x_1) \cos(j\pi x_2)$$

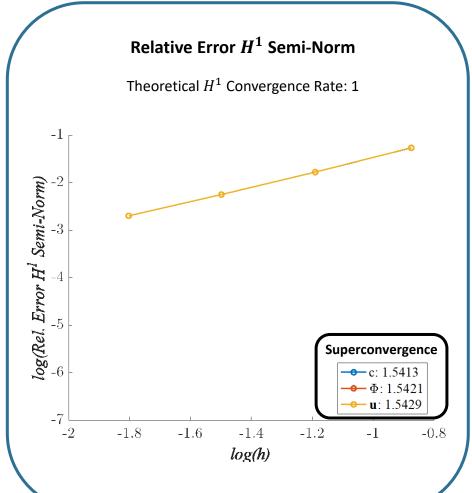
$$\Phi^p = b_0 + A^{\Phi} \cos(i\pi x_1) \cos(j\pi x_2)$$

$$u_1^p = \frac{\beta_1 A^c}{i\pi} \sin(i\pi x_1) \cos(j\pi x_2)$$

$$u_2^p = \frac{\beta_2 A^c}{i\pi} sin(i\pi x_1) \cos(j\pi x_2)$$

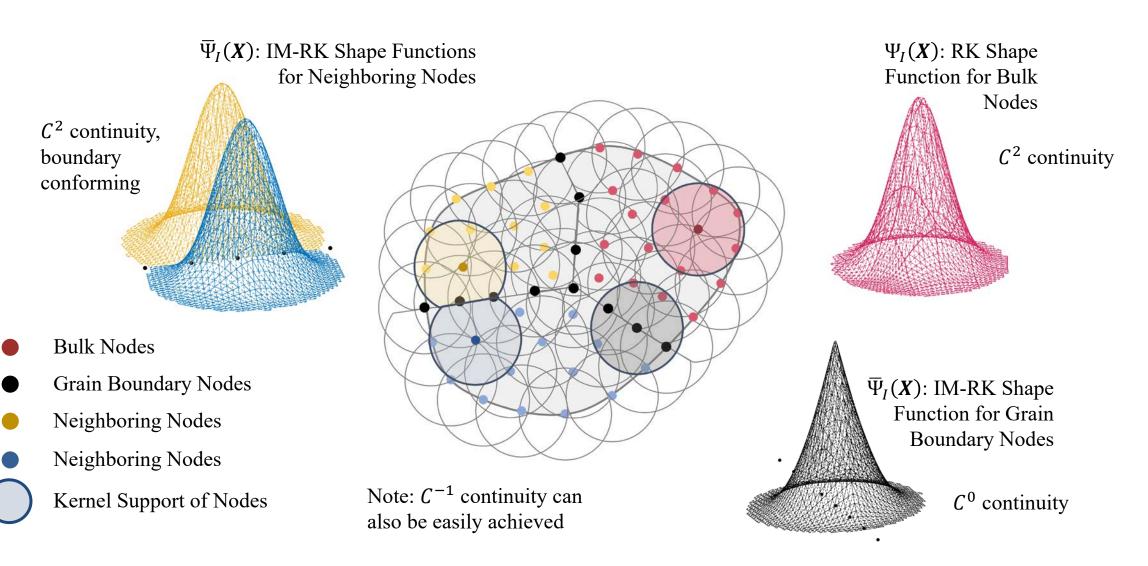
Frequency: i = j = 2





Interface-Modified RK (IM-RK) Approximation for Weak and Strong Discontinuities

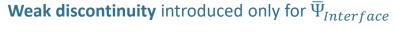
Interface-Modified RK (IM-RK) Approximation for Weak and Strong Discontinuities

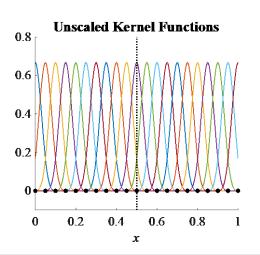


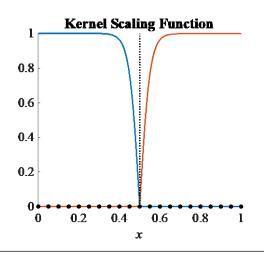
modeling of microstructures", Computational Mechanics.

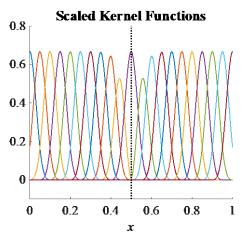
Kernel Function Modifications for Grain Boundaries: max[tanh(dist), 0]

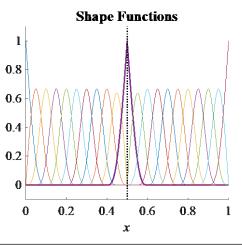
IM-RK with Weak Discontinuity: Scaling with node on interface





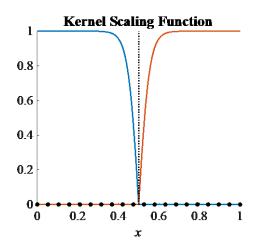




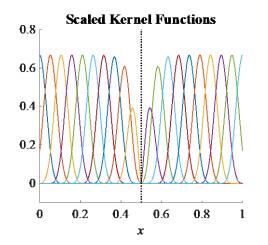


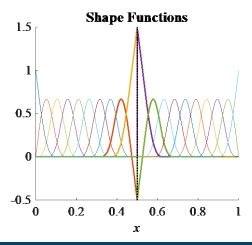
IM-RK with Strong Discontinuity: Scaling with no node on interface

0.6 0.4 0.2 0 0.2 0.4 0.6 0.8 1



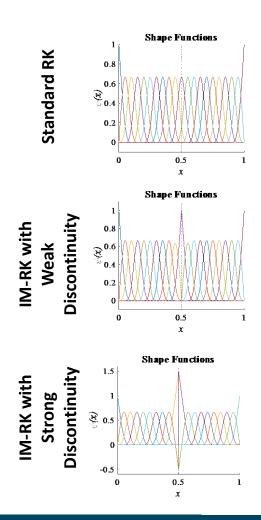
Strong discontinuity introduced only for $\overline{\Psi}_{Interface}$

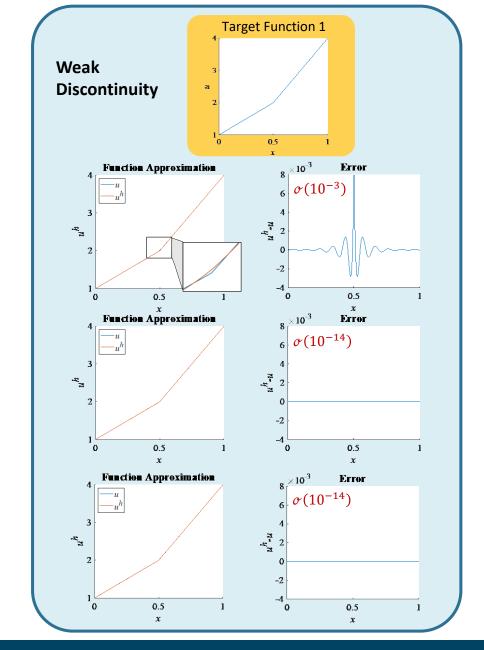




^{9.} **Susuki, K**., J. Allen, J.S. Chen. 2024. "Image-based Modeling of Coupled Electro-Chemo-Mechanical Behavior of Li-ion Battery Cathode Using an Interface-Modified Reproducing Kernel Particle Method." *Eng Comput* (Submitted)

Function Approximation, u^h





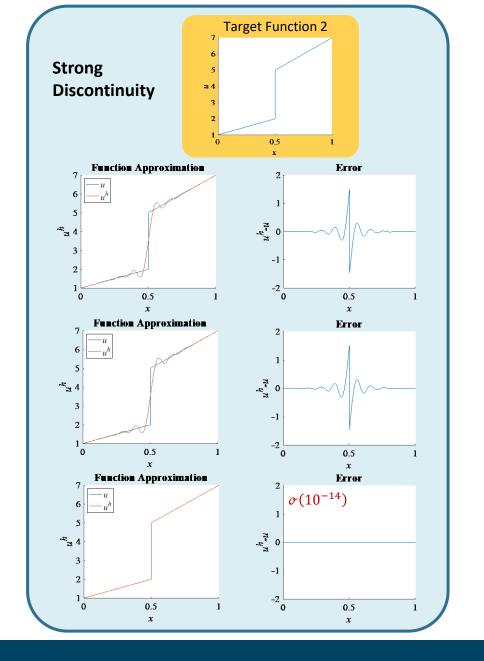
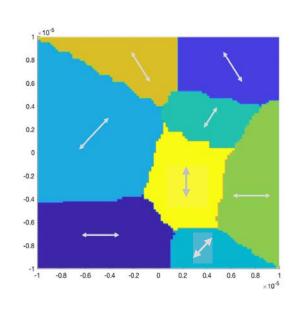
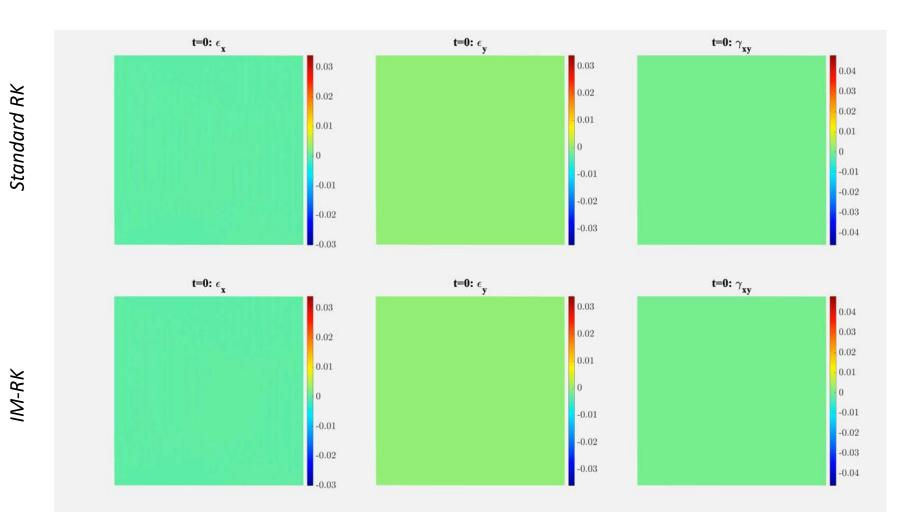
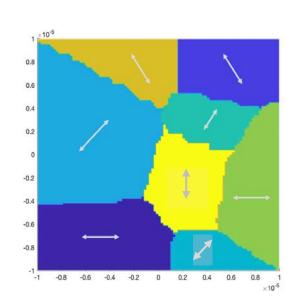


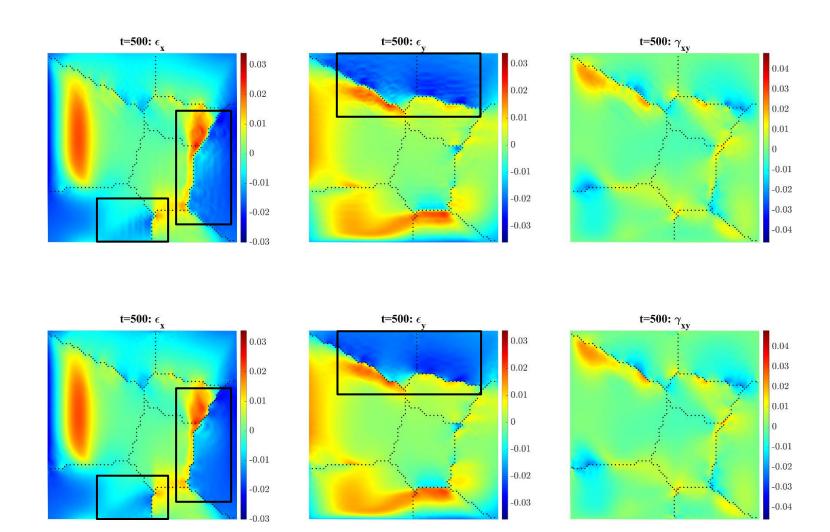
Image-Based Modeling of Statistically-Driven Li-ion Battery Microstructures





^{9.} **Susuki, K**., J. Allen, J.S. Chen. 2024. "Image-based Modeling of Coupled Electro-Chemo-Mechanical Behavior of Li-ion Battery Cathode Using an Interface-Modified Reproducing Kernel Particle Method." Eng Comput (Submitted)





Standard RK

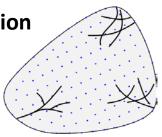
IM-RK

Neural Network-Enhanced RKPM

Neural Network Enhanced Reproducing Kernel (NN-RK) Approximation

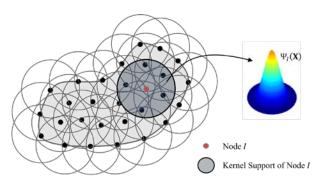
Solution decomposition

$$\mathbf{u}^h = \widetilde{\mathbf{u}}^h + \widehat{\mathbf{u}}^h$$



Smooth solution approximation

$$\widetilde{\mathbf{u}}^h(\mathbf{X}) \approx \mathbf{u}^{RK}(\mathbf{X}) = \sum_{I=1}^{NP} \Psi_I(\mathbf{X}) \mathbf{d}_I$$



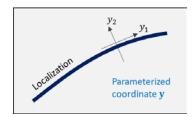
Neural Network (NN) Enrichment

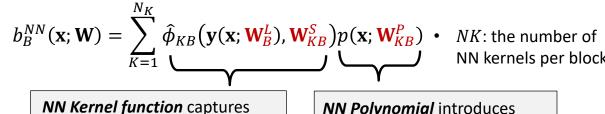
$$\widehat{\mathbf{u}}^h(\mathbf{x}) \approx \mathbf{u}^{NN}(\mathbf{X}) = \sum_{I=1}^{NB} b_I(\mathbf{X}; \mathbf{W})$$

Neural network (NN) approximation

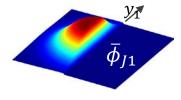
$$u^{NN}(\mathbf{x}) = \sum_{B=1}^{N_B} b_B^{NN}(\mathbf{x}; \mathbf{W}_B)$$
 • b_B^{NN} : block-level NN approximation

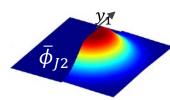
Block-level NN approximation





- Location and orientation of localization
- Shape of solution transition



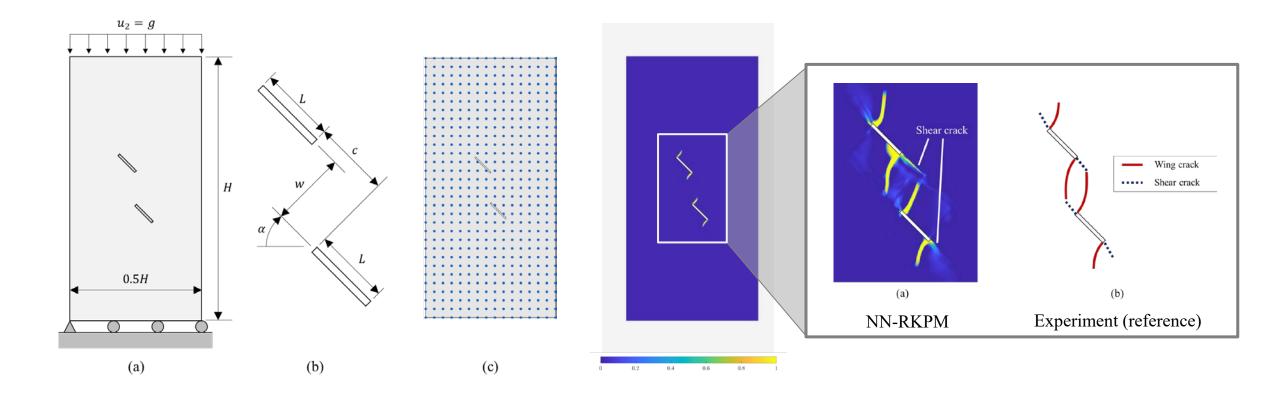


- W^L: NN weight set controlling the location and orientation of the kernel.
- W^S: NN weight set controlling the shape of transition.

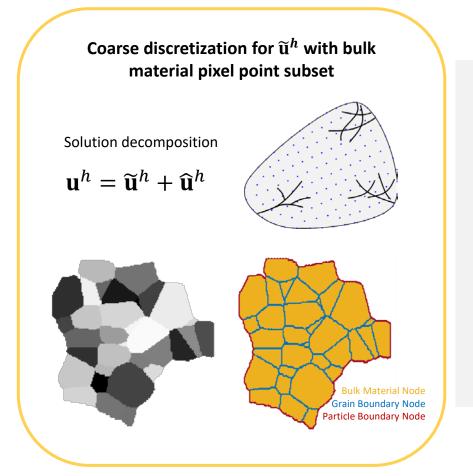
- **NN Polynomial** introduces
- Monomial completeness for further accuracy
- **W**^P: NN monomial coefficient set
- * The NN control parameters \mathbf{W}^L , \mathbf{W}^S , and \mathbf{W}^P are **automatically** determined via loss function minimization.

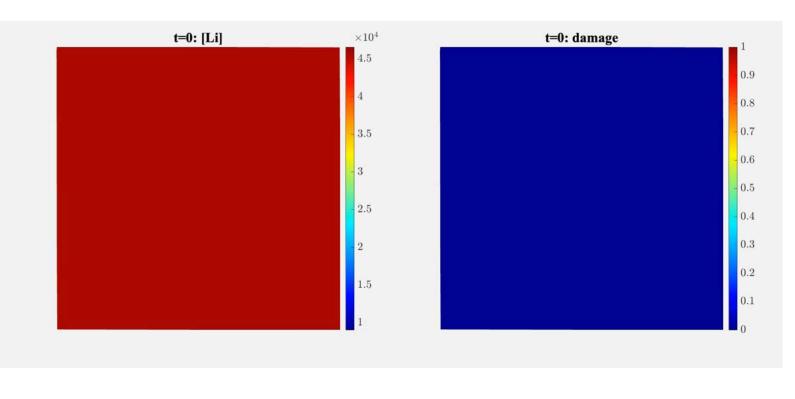
NN kernels per block

Mixed-mode Fracture of Doubly Notched Crack Branching in Isotropic Media



Future Work: Approach for NNRK with Heterogeneous Media under Multi-Physics Loading





Conclusions

- A coupled linear patch test has been formulated and passed for the coupled electro-chemomechanical system, and optimal convergence rates are achieved.
- The interface-modified RK (IM-RK) approximation can introduce various discontinuities by leveraging kernel scaling and strategic interface node placement.
- IM-RK discontinuity introduction shows significant Gibbs oscillation reduction without additional degrees of freedom.
- NNRK is designed to be computationally efficient by superimposing a coarse solution with a localized NN enrichment for fine/localized features.
- NN block-level approximations are designed to capture low order topology but can be superimposed to capture complex topological geometries.

Thank you

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